

Nanostructured Hybrid Shells of r-GO/AuNP/*m*-TiO₂ as Highly Active Photocatalysts

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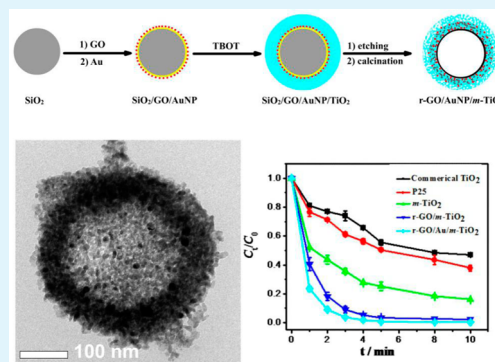
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S Supporting Information

ABSTRACT: Nanostructured hybrid shells of r-GO/AuNP/*m*-TiO₂ were synthesized using SiO₂ spheres as templates, followed by graphene oxide (GO) and Au nanoparticle (AuNP) deposition and TiO₂ coating, and then post-treatments of template removal and calcination. Evaluation of their photocatalytic activity by degradation of Rhodamine B (RhB) under the irradiation of UV, visible light, and simulated daylight demonstrated the superior photocatalytic performance of the sandwich-like hollow hybrid shells, which could be attributed to the porous nature of the hybrid shells and the enhanced charge separation and visible-light absorption of r-GO and AuNPs.

KEYWORDS: TiO₂, graphene, Au nanoparticle, mesoporous, hybrid shells, photocatalyst



1. INTRODUCTION

Since the discovery of its catalytic activity toward hydrogen production from water under UV light, TiO₂ has been widely studied as the most promising photocatalyst for water splitting and environmental remediation.^{1–4} As a photocatalyst, TiO₂ offers a number of advantageous features, including its relatively high photocatalytic efficiency, excellent chemical stability, low toxicity, and low cost.^{5,6} The development of nanoscale TiO₂ photocatalysts with suitable morphologies, such as nanorods,^{7,8} nanotubes,⁹ and hollow spheres,^{10,11} mainly for increasing the specific surface area and enhancing the exciton transfer rate, has attracted extensive attention. Recent research has established that mesoporous hollow TiO₂ nanostructures display enhanced performance in photocatalysis because of their low density, high surface area, surface permeability, better light-trapping effect, porous characteristics, and organic adsorption.^{12–16} However, the inferior utilization efficiency of visible light limits the practical photocatalytic applications of TiO₂ nanostructures due to the large band gap energy and high recombination rate of electron–hole pairs. One of the potential solutions to improve its efficiency is to shift its absorption from the UV to visible region in order to make better use of both UV and visible radiation. Various strategies, including metal ion implantation,¹⁷ nonmetal doping,¹⁸ and compositing with other semiconductors¹⁹ have been extensively explored to enhance the photocatalytic activity of TiO₂ under visible light. Among them, noble metal nanoparticles such as Au and Ag have been frequently used to enhance the photocatalytic activity in visible light by taking advantage of their unique plasmonic proper-

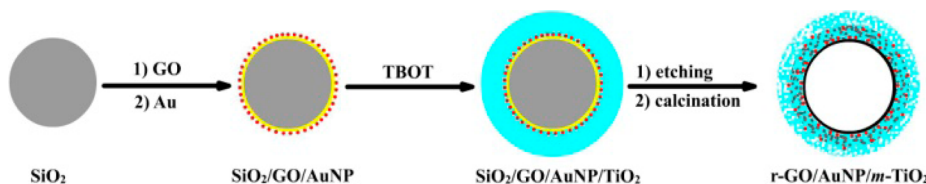
ties.²⁰ In any case, improved absorption of photons may not necessarily guarantee significantly better photocatalytic performance because the efficiency of a photocatalyst is also determined by charge separation and transport.²¹ Graphene nanosheets, a monolayer of carbon atoms arranged in a honeycomb network, have also been reported to hybridize with TiO₂ to slow down the recombination of photo- or electrochemically generated electron–hole pairs, enhance charge transfer rate, and increase surface adsorption of molecules to be decomposed. Generally, the enhancement of photocatalytic performance brought by graphene is attributed to its high specific surface area and superior electron mobility, which make it an ideal sink for photogenerated electrons.^{22–25} It is therefore desirable to design hybrid nanostructures combining TiO₂, plasmonic metal nanoparticles, and graphene, and further explore the possible synergistic effects that may enable superior photocatalytic performances.^{26,27} However, mesoporous hollow TiO₂-based ternary hybrids comprising GO and plasmonic metal nanoparticles with superior photocatalytic performances have been seldom seen.

Herein, we report the successful synthesis of sandwich-structured hybrid nanoshells containing reduced graphene oxide (r-GO), Au nanoparticles (AuNPs), and mesoporous titania (*m*-TiO₂) and demonstrate their high photocatalytic activity toward the decomposition of organic species. In this

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Scheme 1. Schematic Illustration of the Synthesis Steps for r-GO/AuNP/*m*-TiO₂ Sandwich-Like Hollow Spheres

simple and scalable sol–gel based synthesis (Scheme 1), surface-modified SiO₂ spheres were used as templates, loaded sequentially with GO and AuNPs, and finally covered with a layer of TiO₂. Post-treatments including template removal and calcination led to the formation of sandwich-structured r-GO/AuNP/*m*-TiO₂ hollow shells, which possessed high surface area and exhibited excellent adsorption capability. The superior photocatalytic performance of the hybrid nanostructures was fully demonstrated in degradation of Rhodamine B (RhB) under UV light, visible light, and simulated daylight irradiations, suggesting their intriguing potential as effective photocatalysts.

2. EXPERIMENTAL METHODS

2.1. Materials. All reagents were directly used as received without further purification. Tetraethyl orthosilicate (TEOS, 99.8%), P25 (20% rutile and 80% anatase), commercial TiO₂ (99–100.5%), and 3-aminopropyltriethoxysilane (APTES) (99%) were obtained from Sigma-Aldrich. Tetrabutyl orthotitanate (TOBT, 97%) was obtained from Fluka. Anhydrous ethanol, 2-propanol (99.7%), trisodium citrate dihydrate (99%), RhB, MB, and MO, graphite powder (99.85%), KMnO₄ (99.5%), H₃PO₄ (85%), and all other reagents were obtained from Sinopharm Chemical Reagent Co. Ltd. (China). The water used in this study was deionized by milli-Q Plus system (Millipore, France), showing 18.2 MΩ electrical resistivity.

2.2. Synthesis of GO. Typically, 0.3 g graphite and 1.8 g KMnO₄ were added into a mixture of 4 mL H₃PO₄ and 36 mL of concentrated H₂SO₄. After being heated to 50 °C and kept stirring for 12 h, 40 mL of ice containing 300 μL H₂O₂ was slowly added into the mixture. Then, the sample was washed with water, 3.4% HCl (aq), and ethanol by centrifugation, and then redispersed in ethanol at a concentration of 0.5 mg/mL under sonication for 24 h.

2.3. Synthesis of AuNPs. Typically, 0.24 mL of HAuCl₄ (0.1 mol/L) was added to 20 mL of water at room temperature (20–23 °C). After 1 min of stirring, 2.00 mL of 38.8 mmol/L sodium citrate was added. One minute later, 1.00 mL of fresh 0.075% NaBH₄ in 38.8 mmol/L sodium citrate was added. The colloidal solution (0.01 mol/L) was stirred for an additional 5 min and stored at 4 °C in the dark.

2.4. Synthesis and Surface Modification of SiO₂ Spheres. In a typical synthesis, 3.44 mL of TEOS was injected into a mixture of 17.2 mL of water, 2.48 mL of ammonia, and 92 mL of ethanol at room temperature under magnetic stirring. After reacting for 4 h, the colloidal spheres were collected by centrifugation and then redispersed in 10 mL ethanol. After washing twice with ethanol by centrifugation, the particles were transferred to a mixture containing isopropanol (40 mL) and APTES (300 μL) and refluxed for 3 h to functionalize the silica surface with –NH₂ groups. The surface modified particles were washed with ethanol and then redispersed in ethanol (2.5 mg/mL).

2.5. Synthesis of r-GO/AuNP/*m*-TiO₂ Hollow Spheres. The SiO₂/GO hybrids were fabricated simply by mixing APTES-modified SiO₂ ethanol solution and GO ethanol solution. Typically, 0.40 mL GO ethanol solution (0.5 mg/mL) was added into a 40 mL APTES-modified SiO₂ ethanol solution (2.5 mg/mL) under magnetic stirring and then the mixed solution was refluxed for 1 h. After that, SiO₂/GO hybrids were centrifuged and redispersed in 20 mL aqueous solution for further use. The SiO₂/GO/AuNP hybrids were prepared by mixing AuNP aqueous solution (1.2 mL, 0.01 mol/L) with 20 mL SiO₂/GO aqueous solution under sonication for 1 h. SiO₂/GO/AuNP hybrids were then centrifuged and washed twice with ethanol and then

redispersed in 40 mL ethanol. The outer layer of TiO₂ was prepared by TBOT hydrolysis. Typically, 40 mL SiO₂/GO/AuNP ethanol solution was mixed with 0.4 mL TBOT under vigorous magnetic stirring for 1 h. Then, 20 mL water was injected into the mixture using a syringe pump at a rate of 0.5 mL/min, followed by magnetic stirring for 12 h. The as-formed SiO₂/GO/AuNP/TiO₂ hybrids were washed twice with water, redispersed in 30 mL of water under ultrasonication for 30 min, and then etched with NaOH solution (10 mL, 2.5 mol/L) at the desired temperature (90 °C) for 3 h to completely remove SiO₂ cores. After that, the solution was centrifuged and redispersed in 20 mL HCl solution (0.1 mol/L) under magnetic stirring for 1 h. Then the solution was washed twice with water and dried under vacuum. Finally, GO/AuNP/TiO₂ hybrids were calcined at 600 °C under N₂ protection for 4 h to crystallize the amorphous TiO₂ to small anatase TiO₂ grains and to change GO into r-GO through pyrolysis. The obtained r-GO/AuNP/*m*-TiO₂ hybrid hollow shells were further used as photocatalysts.

2.6. Catalytic Experiments. The photocatalytic activity was tested by degradation of RhB. Before the reaction was initiated, the catalysts were first irradiated under UV light with wavelength of 365 nm for 30 min to remove any residual organic contaminants. The catalyst (5 mg) was added to an aqueous RhB solution (25 mL, 2.0 × 10^{−5} mol/L) in a 50 mL quartz cell and the solution (pH = 7) was stirred in the dark for 30 min to ensure equilibrium adsorption of the dye. A UV lamp (300 W Hg) with a 365 nm filter, a visible lamp (400 W Metal Halide) with a 400 nm filter, and a sunlight simulator (350 W xenon) were used in a commercial photoreactor system (Xujiang XPA-7), the focused intensity values on the flask were 1.75 mW, 3.90 mW, and 2.01 mW, respectively. It was measured using a light dynamometer (CEL-NP2000, CEAULIGHT) with the range of 0–2000 mW. The concentration of RhB before and after irradiation was determined by measuring the absorbance at 553 nm with a UV–vis spectrophotometry (HR2000CG-UV-NIR, Ocean Optics).

2.8. Instruments. The sample morphology was characterized using transmission electron microscopy (TEM, Tecnai T12 Philip Apparatus Co., United States) and high resolution transmission electron microscopy (HRTEM, Tecnai G2 F30 S-Twin TEM, FEI). The crystal phase was evaluated by X-ray diffraction (XRD) analysis using a Bruker AXS D8 ADVANCE X-ray diffractometer. The nitrogen adsorption isotherm was obtained at 77 K using a Beishide 3H-2000PS2 specific surface and pore size analysis instrument. The surface area was calculated from the adsorption isotherm using the multipoint Brunauer–Emmett–Teller (BET) method in the pressure range of $P/P_0 = 0.05–0.25$. The average pore sizes were determined by the Barrett–Joyner–Halenda (BJH) method from the adsorption isotherm. The optical properties were tested by UV/vis diffuse reflectance spectra (Cary 5000, Varian). X-ray photoelectron spectroscopy (XPS) data were recorded on a PHI Quantera II using a nonmonochromatized Al Kα X-ray (1486.6 eV) as the excitation source and choosing C 1s as the reference line. The electrochemical impedance spectra (EIS) were carried out on an electrochemical workstation (PGSTAT30, Auto Lab) by using three-electrode cells. The resultant electrode served as the working electrode, with a platinum wire as the counter electrode, and a Ag/AgCl (saturated KCl) electrode as the reference electrode. The measurements were performed in the presence of a 0.5 mol/L Na₂SO₄ solution.

3. RESULTS AND DISCUSSION

3.1. Morphology and Formation Mechanism of r-GO/AuNP/*m*-TiO₂ Hollow Spheres. Scheme 1 illustrates the synthesis steps involved in the preparation of r-GO/AuNP/*m*-TiO₂ sandwich-like hollow shells. Monodispersed SiO₂ particles were chosen as templates, followed by the self-assembly of GO on SiO₂ surfaces and further decoration of AuNPs on GO surfaces. Finally, TiO₂ layers were coated on SiO₂/GO/Au hybrid surfaces, leading to r-GO/AuNP/*m*-TiO₂ sandwich-like hollow spheres after template removal and calcination.

The apparent color changes of colloidal samples during synthesis were monitored as given in Figure S1. To ensure effective coating of GO, SiO₂ particles have been modified with a coupling agent APTES. As driven by electrostatic interactions, GO can be uniformly self-assembled onto surfaces of SiO₂ particles (Figure S2). The SEM and TEM images as given in Figure 1a and b clearly reveal the formation of SiO₂/GO core/

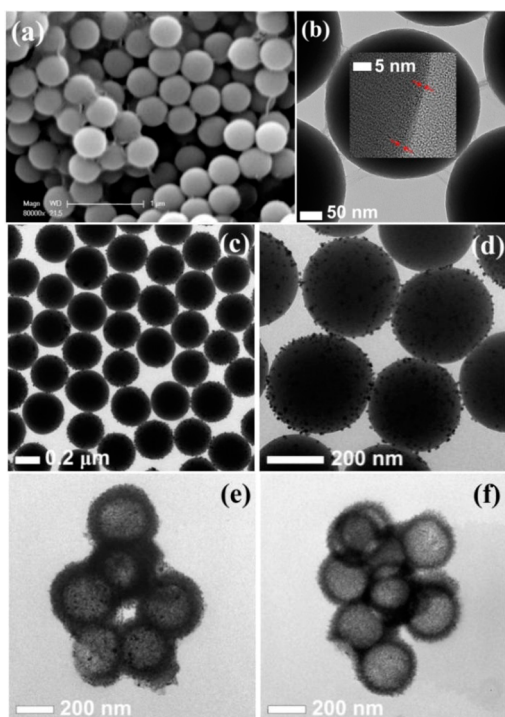


Figure 1. (a) SEM and (b) TEM images of SiO₂/GO core/shell hybrids. (c, d) TEM images of SiO₂/GO/AuNP hybrids. (e) TEM image of r-GO/AuNP/*m*-TiO₂ sandwich-like hollow shells. (f) TEM image of *m*-TiO₂ hollow shells.

shell hybrids. The high-resolution TEM (HRTEM) image in Figure 1b shows two or three layered GO nanosheets on surfaces of SiO₂. The subsequent step is to introduce AuNPs. There are two ways to realize the decoration of AuNPs on surfaces of SiO₂/GO hybrids: (1) the addition of presynthesized Au colloids, which can be located on surfaces of SiO₂/GO. GO, which contains carboxylic groups and π electrons, can interact with AuNPs (in addition, the remaining amino groups of modified SiO₂ may also cause adsorption of AuNPs on surfaces of SiO₂/GO); (2) the addition of HAuCl₄, which can result in the formation of AuNPs on the surface of GO after redox reactions between GO and AuCl₄⁻.²⁸ Although the latter strategy shows the obvious advantage of the simplified procedure, the size and loading amount of in situ formed AuNPs are hard to control, which makes it difficult to reveal the

effect of AuNPs on their catalytic performances. Therefore, we choose the former one to introduce AuNPs with predetermined size and amount. AuNPs of 5 nm in diameter were adsorbed onto the GO surface by strong chemical affinities, as determined by the color change of the original Au suspension from red to colorless under controlled centrifugation to isolate only SiO₂/GO particles. The formation of SiO₂/GO/AuNP hybrids has been further confirmed by TEM images (Figure 1c and d). Afterward, the outer amorphous TiO₂ shells have been formed through hydrolysis of TBOT. Finally, the SiO₂ cores have been thoroughly removed through the well-established etching process using concentrated NaOH solution,^{14–16} and then the calcination process under N₂ ambient conditions to ensure crystallization and mesoporous shell formation of TiO₂ and the formation of r-GO derived from GO through pyrolysis.²⁹ As given in Figure 1e, r-GO/AuNP/*m*-TiO₂ hollow spheres can be clearly evidenced. In comparison with *m*-TiO₂ hollow spheres without GO and AuNPs synthesized using a similar strategy (Figure 1f), the dark dots mainly located in the inner surfaces of r-GO/AuNP/*m*-TiO₂ (Figure 1e) should be ascribed to AuNPs.

In order to fully understand the sandwich-structured r-GO/AuNP/*m*-TiO₂ hollow shells, the low- and high-magnification TEM images are given in Figure 2a and b, respectively. The inner diameter is similar to the size of SiO₂ template and the mesoporous TiO₂ shell is around 50 nm in thickness. Figure 2c depicts the HRTEM image of r-GO/AuNP/*m*-TiO₂ hollow spheres. The observed fringe spacing of 0.35 nm ensures the presence of anatase TiO₂ (101) plane (JCPDS 21–1272), whereas the spacing of 0.236 nm indicates the presence of Au (111) plane (JCPDS 04–0784). The high-angle annular dark field scanning transmission electron microscopy (HAADF-STEM) image of a single r-GO/AuNP/*m*-TiO₂ particle as given in Figure 2d clearly reveals the location of AuNPs mostly on the inner surface of the TiO₂ shell. However, some AuNPs penetrating into TiO₂ porous shells can also be seen. In addition, the energy dispersive X-ray spectroscopic (EDS) elemental maps of Ti (Figure 2e), Au (Figure 2f), and C (Figure 2g) in a single shell in Figure 2d further confirm the expected structure. It should be noted that both Au and C partially penetrated into TiO₂ porous shells can be verified. As the post-treatment of calcination was applied to crystallize TiO₂ from amorphous into crystalline structure, the penetration of AuNPs and GO into TiO₂ shell is possibly related to the porous shell formation with voids together with high temperature induced migration. It was also found that AuNPs remained the same size even after harsh etching and calcination treatment. In a control experiment, AuNP/*m*-TiO₂ hollow spheres were synthesized under similar conditions without the introduction of GO. It was clearly seen that AuNPs aggregated to form large particles (Figure S3). As a result, it is concluded that the presence of GO can effectively prohibit the aggregation of AuNPs during etching and calcination processes possibly due to its strong affinity toward AuNPs.

3.2. Characterization of r-GO/AuNP/*m*-TiO₂ Hollow Spheres. From the XRD pattern of r-GO/AuNP/*m*-TiO₂ hollow hybrids as shown in Figure 3a, all the identified peaks for TiO₂ were assigned to anatase (JCPDS, no. 73–1764). The average crystal size of TiO₂ in r-GO/AuNP/*m*-TiO₂ hollow hybrids is approximately 3 nm as calculated from the most intensive (101) peaks based on the Scherrer formula. Besides, additional peaks corresponding to Au have also been observed. Note that a broad diffraction peak at around 26° corresponding

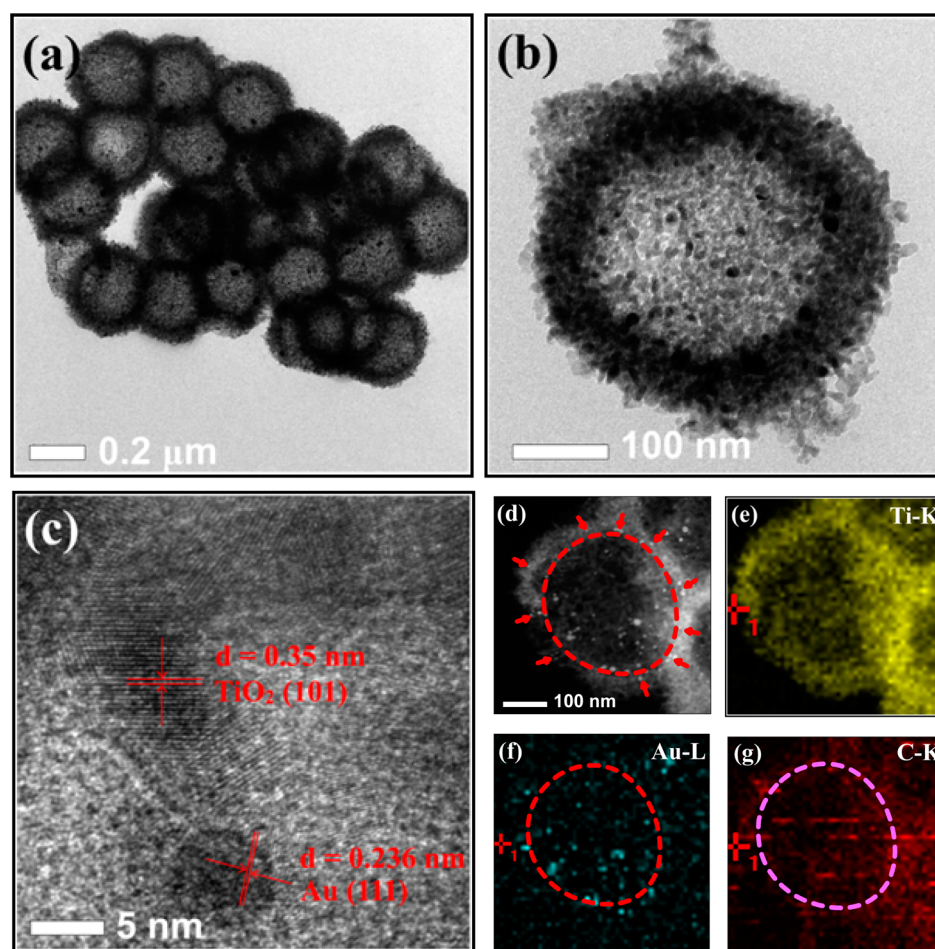


Figure 2. (a, b) TEM and (c) HRTEM images of r-GO/AuNP/*m*-TiO₂ hollow spheres. (d) HAADF-STEM image of r-GO/AuNP/*m*-TiO₂ hollow spheres and (e, f) EDS maps of (e) Ti, (f) Au, and (g) C from a single particle given in Figure 1d.

to r-GO was not observed in the XRD pattern due to the strong diffraction from crystalline TiO₂. In the XPS spectrum of r-GO/AuNP/*m*-TiO₂ hollow hybrids (Figure S4), only the signatures of Ti, O for TiO₂, and Au, C for Au and r-GO can be evidenced. The XPS signature of Au 4f doublet peaks (Figure 3b) for AuNPs locate at 82.3 and 86.0 eV with the splitting of the 4f doublet 3.7 eV, suggesting the metallic nature of Au. However, the peaks show a large shift to the direction of low energy according to the XPS handbooks and the previous reports.³⁰ For comparison, XPS spectra of GO/AuNP/TiO₂ hollow hybrids (before calculation for r-GO/AuNP/*m*-TiO₂ hollow hybrids) were also recorded, where the Au 4f doublet peaks for AuNPs in GO/AuNP/TiO₂ hollow hybrids were located at 83.8 and 87.5 eV. In addition, the Ti 2p doublet peaks for TiO₂ in GO/AuNP/TiO₂ are located at 459.0 and 464.8 eV, whereas those in r-GO/AuNP/*m*-TiO₂ hollow hybrids locate at 457.8 and 463.5 eV, both with the splitting of the 2p doublet 5.7 eV (Figure 3c), suggesting a normal state of Ti⁴⁺. In addition, the Au 4f and Ti 2p binding energy shows 1.3 and 1.2 eV shifts, respectively, both in the direction of low energy. Figure 3d shows the C 1s peaks of GO and r-GO/AuNP/*m*-TiO₂ hollow hybrids. A high ratio of nonoxygenated C to the carbonyl C in r-GO/AuNP/*m*-TiO₂ hollow hybrids can be evidenced, suggesting a considerable degree of reduction for the GO nanosheets.³¹ The shift of Au 4f indicates that some electrons may migrate from TiO₂ to metallic Au and there is a strong interaction between AuNPs and TiO₂. Because AuNPs

can act as electron acceptors and help to separate the photoexcited electron–hole pairs, the hybrid structure can inhibit the recombination of excited electrons and holes. The shifts of 1.2 eV for Ti 2p may be attributed to the interactions of Ti with oxygen centers of r-GO. It is noted that a slight shift to lower binding energy is observed for the peaks of the C 1s spectra after the thermal treatment, suggesting the formation of Ti–O–C bonds between TiO₂ and r-GO after high temperature annealing.^{32–35} The above results indicate that strong interactions in the ternary hybrids can be verified.

The surface area and porosity of r-GO/AuNP/*m*-TiO₂ hollow hybrids were measured by N₂ adsorption–desorption isotherm analysis, with the results shown in Figure 3e. The isotherm is of the typical type III pattern with H₃ hysteric loop in the range of 0.5–1.0 *P/P*₀, indicating the existence of slit-shaped pores according to the IUPAC classification. The surface area as measured by the multipoint BET method from the adsorption branch is 171.4 m² g^{−1}, which is much higher than commercial P25 (about 50 m² g^{−1}). The pore size distribution of the sample as determined using the BJH method from the adsorption branch of the isotherm is shown in the inset in Figure 3e, which suggests an average pore diameter of 6.0 nm.

The UV/vis diffuse reflectance spectra were conducted to uncover the optical properties of r-GO/AuNP/*m*-TiO₂ hollow hybrids (Figure 4a). The addition of r-GO induced the increased light absorption intensity in visible light regions, as

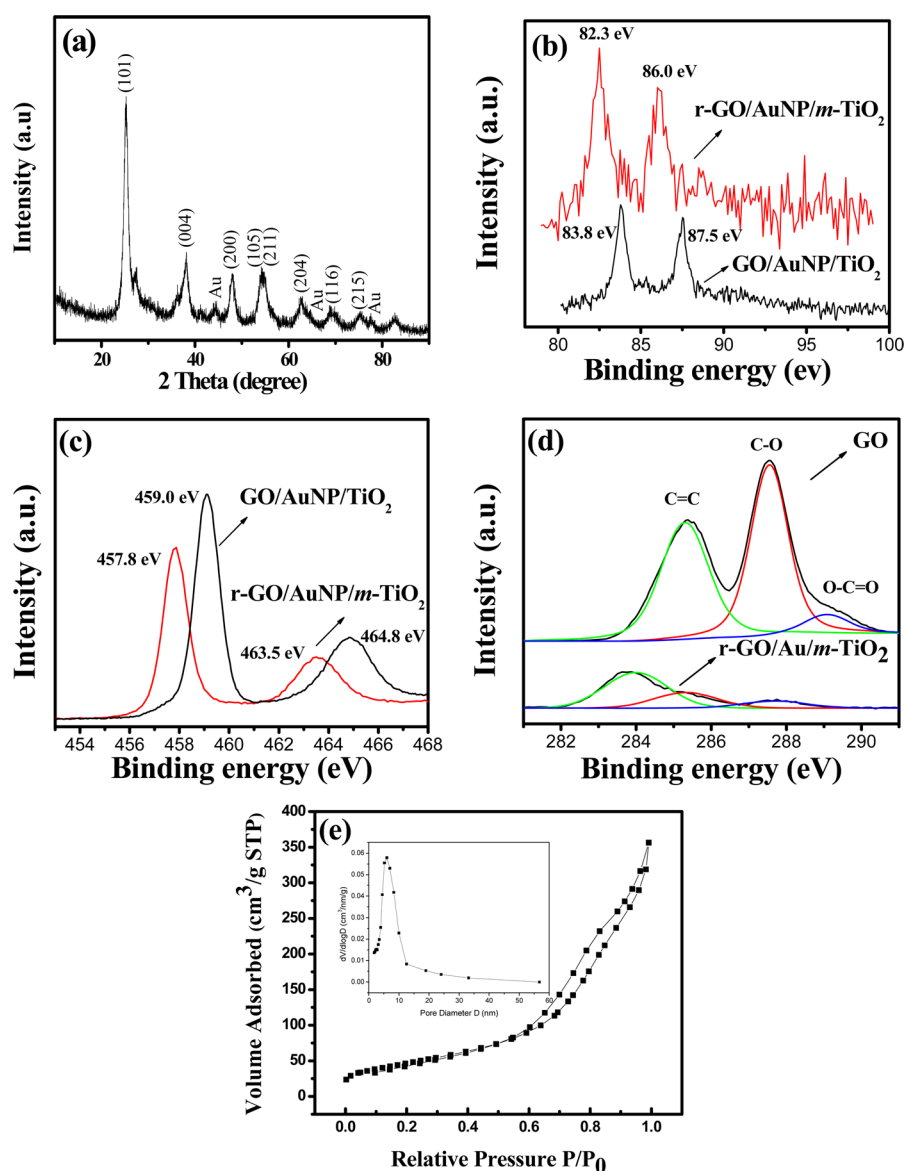


Figure 3. (a) XRD pattern of r-GO/AuNP/m-TiO₂ hollow hybrids. (b–d) XPS spectra of (b) Au 4f, (c) Ti 2p, and (d) C 1s. (e) Nitrogen adsorption isotherm of r-GO/AuNP/m-TiO₂ hollow hybrids. Inset in Figure 2e shows the pore size distribution as calculated by using the BJH method.

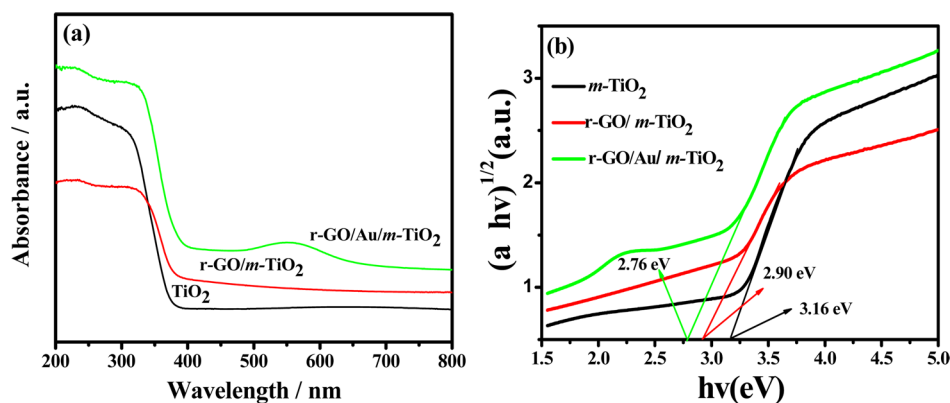


Figure 4. (a) UV/vis diffuse reflectance spectra of *m*-TiO₂, r-GO/*m*-TiO₂, and r-GO/AuNP/*m*-TiO₂ hollow spheres and (b) their corresponding Kubelka–Munk plots.

compared to the bare *m*-TiO₂. Moreover, another absorption band for r-GO/AuNP/*m*-TiO₂ hollow hybrids at around 550

nm appeared, which could be ascribed to the surface plasmon resonance of AuNPs. Furthermore, a qualitative red shift to

longer wavelength is also observed in the absorption edge. Figure 4b shows the plots of the Kubelka–Munk remission function corresponding to each spectrum, which indicates that bare anatase $m\text{-TiO}_2$ was 3.16 eV, while the band gap of the $r\text{-GO}/m\text{-TiO}_2$ estimated from the spectrum was 2.90 eV, and the band gap of the $r\text{-GO}/\text{Au}/m\text{-TiO}_2$ hollow spheres were significantly reduced to 2.76 eV. This phenomenon should occur due to the interaction between graphene and TiO_2 and the surface plasmon resonance of AuNPs.³⁶

On the basis of above results, the morphology and chemical structures of $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$ hollow hybrids can be successfully verified. It is believed that the mesoporous shells with large surface area and enhanced visible absorption may potentially promote their photocatalytic performance.

3.3. Catalytic Activities of $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$ Photocatalysts in the Degradation of RhB. The activity of $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$ photocatalysts was evaluated by monitoring the photodegradation of RhB under different light irradiation. Before light irradiation, the solution containing catalysts and dyes was stirred in the dark for 30 min to ensure equilibrium adsorption of RhB on surfaces of catalysts. As shown in Figure 5, $m\text{-TiO}_2$ hollow spheres show higher adsorptivity than

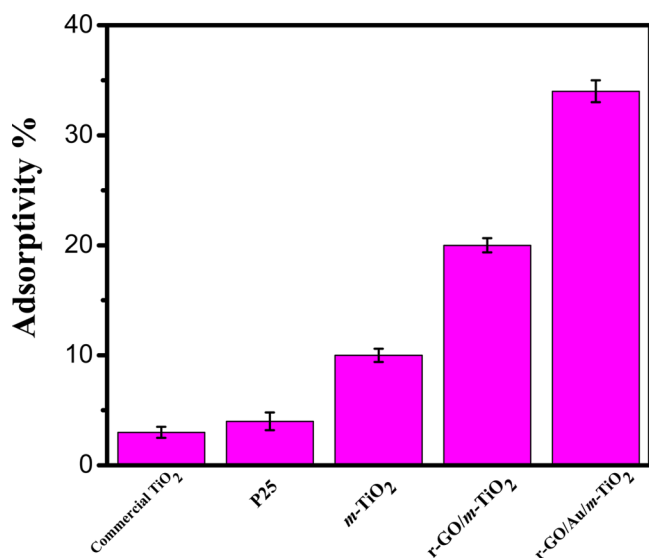


Figure 5. Bar plot showing the adsorptivity of RhB with different photocatalysts.

commercial TiO_2 and P25. In addition, the introduction of $r\text{-GO}$ and AuNPs in mesoporous hollow TiO_2 can significantly increase the adsorptivity of RhB, which can reach as high as 34% for $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$. It is possible that the introduction of AuNPs in $r\text{-GO}/m\text{-TiO}_2$ will create more nanopores between $r\text{-GO}$ and $m\text{-TiO}_2$ in the hybrid shells, leading to better adsorptivity than $r\text{-GO}/m\text{-TiO}_2$.

The performance of $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$ photocatalysts toward the rate of RhB degradation was measured by monitoring the adsorption intensity of the characteristic RhB peak at 553 nm. For comparison, the degradation of RhB with commercial TiO_2 , P25, $m\text{-TiO}_2$ and $r\text{-GO}/m\text{-TiO}_2$ photocatalysts was also measured. Figure 6a shows time profiles of C_t/C_0 under UV irradiation, where C_t is the concentration of RhB at irradiation time t and C_0 the concentration at absorption equilibrium of the photocatalyst before irradiation. It is clear that $m\text{-TiO}_2$ hollow spheres show higher degradation efficiency than commercial TiO_2 and P25 possibly due to their large surface area, superior adsorption ability, and possible light-trapping effect resulting from mesoporous shells. Considering the electric conductivity and visible light absorption of $r\text{-GO}$, it is not surprising that the introduction of $r\text{-GO}$ to $m\text{-TiO}_2$ can enhance the degradation efficiency. More interestingly, the presence of AuNPs in $r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$ could further improve their degradation efficiency with the rate constant reaching as high as 1.02 min^{-1} (Figure 6b), which is believed to surpass most of the reported TiO_2 -based photocatalysts (Table 1). The results undoubtedly confirm the positive impact of $r\text{-GO}$

Table 1. Comparison of the Photocatalytic Activity of Different TiO_2 -Based Photocatalysts

sample	k (min^{-1})	irradiation	organic dye	ref
$\text{TiO}_2\text{-PE-800}$	0.15	UV	RhB	15
TiO_2 hollow spheres	0.013	UV	MB	37
graphene/ TiO_2 mixture	0.20	UV	RhB	38
$\text{H-TiO}_2\text{-800}$	0.20	UV	RhB	16
Au-TiO_2 nanocups	0.022	UV	MB	12
$\text{Fe}_3\text{O}_4/\text{SiO}_2/\text{TiO}_2$	0.17	UV	RhB	39
$\text{TiO}_2/\text{TiO}_2$	0.0034	UV	MO	40
Au@TiO_2 core/shell	0.0087	UV	RhB	41
$m\text{-TiO}_2$	0.32	UV	RhB	this work
$r\text{-GO}/m\text{-TiO}_2$	0.73	UV	RhB	this work
$r\text{-GO}/\text{AuNP}/m\text{-TiO}_2$	1.02	UV	RhB	this work

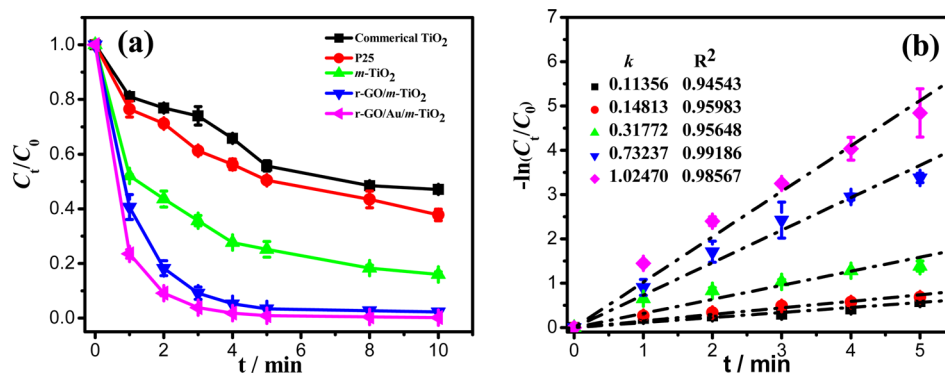


Figure 6. (a) Evolution of RhB concentration versus reaction time under UV irradiation and (b) apparent reaction rate constant versus UV irradiation time in the presence of different photocatalysts.

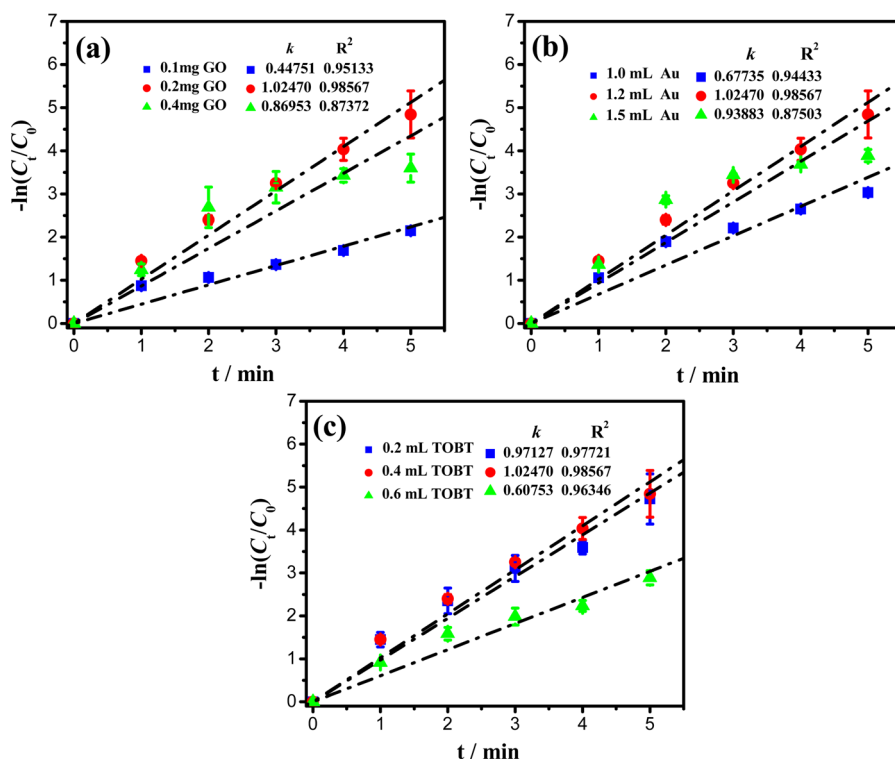


Figure 7. Apparent reaction rate constant versus reaction time under UV irradiation in the presence of r-GO/AuNP/m-TiO₂ with different amount of (a) GO, (b) Au, and (c) TOBT as the photocatalysts.

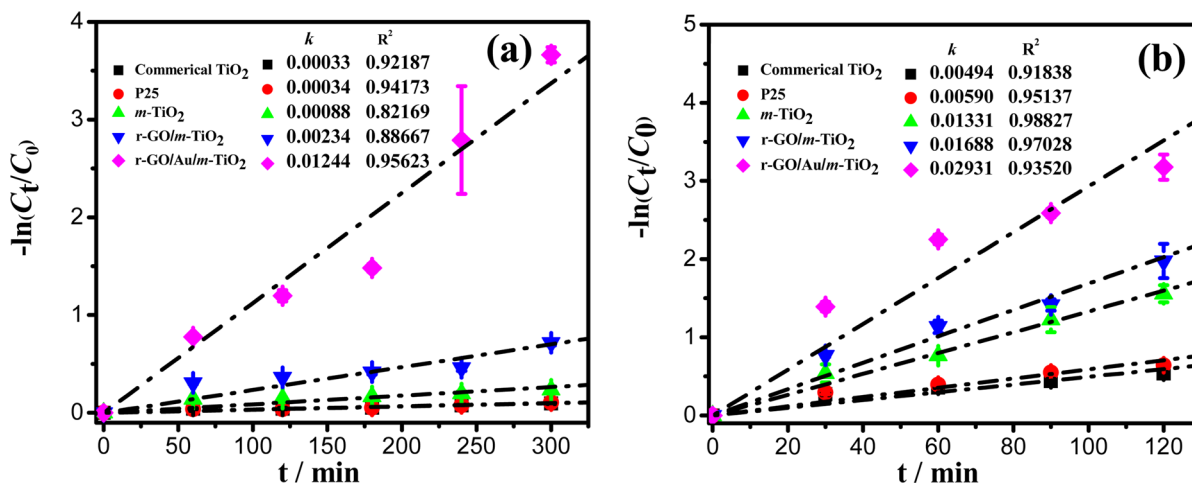


Figure 8. Apparent reaction rate constant versus (a) visible light and (b) simulated daylight irradiation time in the presence of different photocatalysts.

GO and Au decoration on catalytic activity. It is found that r-GO/AuNP/m-TiO₂ photocatalysts are also mechanically robust and chemically stable: they can be recovered and reused to catalyze multiple cycles of degradation reactions under UV light irradiation (Figure S5). In addition, r-GO/AuNP/m-TiO₂ hollow hybrids were also active photocatalysts for degradation of other organic dyes, such as methylene blue (MB) and methyl orange (MO) (Figure S6).

The structures of r-GO/AuNP/m-TiO₂ hollow hybrids were also optimized by varying the compositions to assess the effect of different ingredients in catalytic efficiency. It was found that higher r-GO amount in r-GO/AuNP/m-TiO₂ hollow hybrids led to decreased photocatalytic activity as shown in Figure 7a and Figure S7a, which was consistent with reported

results.^{24,31,42} The increasing coverage of r-GO on TiO₂ with higher r-GO amount will facilitate faster photodegradation due to electric conductivity and visible light absorption of r-GO; however, excessive r-GO not hybridized with TiO₂ should be accounted for the decreased photocatalytic activity.⁴¹ As for AuNP, it is generally understood that AuNPs can help to enhance charge separation by serving as an electron reservoir, and to act as a recombination center which negatively affects catalytic activity.²¹ Therefore, excessive amounts of AuNPs may increase the occurrence of exciton recombination,^{21,43} which leads to decreased photocatalytic activity (Figure 7b and Figure S7b). Figure 7c and Figure S7c reveal the photocatalytic activity variation with TBOT amount. It was found that the TiO₂ shell thickness increased with increasing TBOT amount. The

increase in TiO₂ shell thickness may lead to increased migration distance of photoexcited electrons/holes to the catalyst surfaces, which is believed to diminish the photocatalytic activity. Hereafter, the optimized r-GO/AuNP/*m*-TiO₂ hollow hybrids were chosen as photocatalysts in visible light and simulated daylight irradiation experiments.

For the visible light irradiation experiment, a 400 W metal halide lamp was used as the light source, and a cutoff filter was used to block the UV light (<400 nm). As shown in Figure S8a, r-GO/AuNP/*m*-TiO₂ photocatalysts can decompose more than 97% of RhB within 5 h, while commercial TiO₂ and P25 containing an equivalent amount of TiO₂ can convert only about 10% of RhB under identical photodegradation conditions. The relative photocatalytic activity of the catalysts for RhB degradation follows the order r-GO/AuNP/*m*-TiO₂ > r-GO/*m*-TiO₂ > *m*-TiO₂ > P25 (Figure 8a), which is similar to the trends under conditions of UV light irradiation. A 350 W xenon lamp was used as the light source to simulate daylight irradiation. As shown in Figure S8b, approximately 96.4% of RhB was degraded within 2 h by r-GO/AuNP/*m*-TiO₂ photocatalysts; the value can reach as high as 99.7% when the irradiation time is prolonged to 2.5 h. The relative photocatalytic activity of the catalysts for RhB degradation follows the same order as in the cases of UV and visible light irradiation (Figure 8b).

From the above results, we can conclude that the introduction of r-GO and AuNPs can significantly enhance the photocatalytic performances of *m*-TiO₂ in degradation of RhB under UV, visible light, and simulated daylight conditions. It is known that typical photocatalytic decomposition of organic pollutant species involves three main steps: adsorption of the organic molecules, absorption of light by the photocatalyst, and charge transfer reactions to create the oxidative radical species to decompose the pollutants. Upon irradiation of r-GO/AuNP/*m*-TiO₂ with UV light, electrons are excited from the VB to the CB of TiO₂ generating holes in the VB. Photoexcited electrons and holes can then react with H₂O and/or dissolved oxygen to create oxidative radicals which can undergo chain reactions to decompose the organic pollutants into small molecules.⁴⁴ It is known that the fast intrinsic electron/hole pair recombination results in low catalytic activity. The incorporation of graphene in fact as a bridge for the electron transfer, which largely enhances the transition probability of the valence in TiO₂ to its conduction band, resulting in an increase in photoconduction that is conducive to reduction of the photogenerated carrier recombination rate, and the presence of AuNPs is believed to enhance charge separation by serving as an electron reservoir. Under visible light irradiation, AuNPs in hybrid shells help to harvest the visible-light energy through plasmonic excitation in the visible spectrum, and then charge separation is finished by transfer of electrons from AuNPs to CB of TiO₂. At the same time, the CB electrons of the TiO₂ transfer to r-GO as the mobility of these electrons on r-GO sheets is high due to their unique electric property. The r-GO in the hollow hybrids can behave as photoelectron reservoirs,^{32–35} while under simulated daylight irradiation, it may collect both UV and visible light reaction mechanisms: electrons excited from the VB to the CB of TiO₂, together with electrons from AuNPs due to plasmonic excitation are both transferred to r-GO. The electrochemical impedance spectra were presented as Nyquist plots (Figure 9), in which r-GO/AuNP/*m*-TiO₂ showed the smallest semicircle in the middle-frequency region, followed by r-GO/*m*-TiO₂ and *m*-TiO₂. The results indicate reduced charge transfer resistance

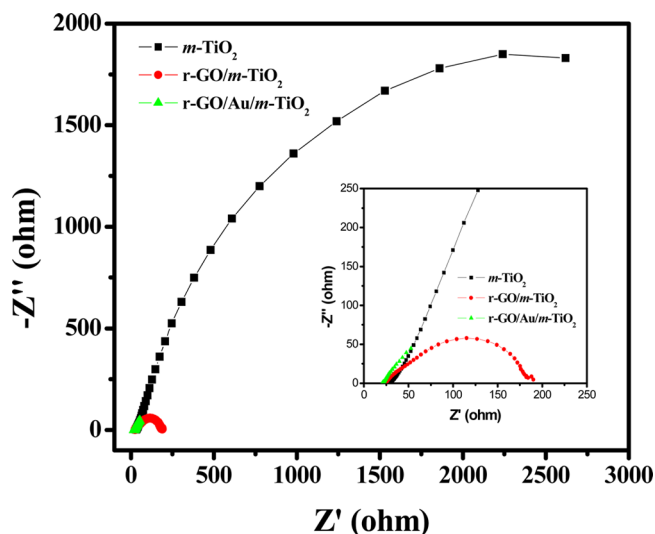


Figure 9. Nyquist plots of *m*-TiO₂, r-GO/*m*-TiO₂, and r-GO/AuNP/*m*-TiO₂ in 0.5 mol/L Na₂SO₄ aqueous solution.

in the solid state interface after the introduction of r-GO,^{45,46} and the addition of AuNPs further deepen the impact. Therefore, the electron accepting and transporting properties of r-GO and AuNPs in r-GO/AuNP/*m*-TiO₂ hollow hybrids can contribute to the suppression of charge recombination, and thereby result in high photocatalytic performance. In addition, the large surface area of r-GO/AuNP/*m*-TiO₂ hollow hybrids with improved dye adsorption may facilitate the transfer of photogenerated excitons to dyes, which also leads to increased photocatalytic activity.

4. CONCLUSION

We report in this work the successful fabrication of sandwich-structured r-GO/AuNP/*m*-TiO₂ hollow hybrid shells with large surface area and superior adsorption ability via the combination of sol-gel and self-assembly methods. Results on photocatalytic degradation of RhB indicated that the introduction of r-GO and AuNPs could improve the catalytic activity of *m*-TiO₂. The superior photocatalytic performance of r-GO/AuNP/*m*-TiO₂ hollow hybrids as revealed by RhB degradation under both UV and visible irradiation clearly suggested their potential application as powerful photocatalysts. We believe the strategy developed here is instructive for construction of other sandwich-structured r-GO/noble metal/*m*-TiO₂ hollow hybrid shells for broader applications such as photocatalysts for water splitting, active layers for hybrid solar cells, and electrode materials for lithium ion batteries.

■ ASSOCIATED CONTENT

Supporting Information

Additional figures giving detailed material characterizations. This material is available free of charge via the Internet at <http://pubs.acs.org>.

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Notes

The authors declare no competing financial interest.

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